

Degradation of AlGaIn/GaN HEMTs under elevated temperature lifetesting

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Abstract

Elevated temperature lifetesting was performed on 0.25 μm AlGaIn/GaN HEMTs grown by MOCVD on 2-in. SiC substrates. A temperature step stress (starting at T_a of 150 °C with a step of 15 °C; ending at T_a of 240 °C; 48 h for each temperature cycle) was employed for the quick reliability evaluation of AlGaIn/GaN HEMTs. It was found that the degradation of AlGaIn/GaN HEMTs was initiated at ambient temperature of 195 °C. The degradation characteristics consist of a decrease of drain current and transconductance, and an increase of channel-on-resistance. However, there is no noticeable degradation of the gate diode (ideality factor, barrier height, and reverse gate leakage current). The FIB/STEM technique was used to examine the degraded devices. There is no detectable ohmic metal or gate metal inter-diffusion into the epitaxial materials. Accordingly, the degradation mechanism of AlGaIn/GaN HEMTs under elevated temperature lifetesting differs from that observed in GaAs and/or InP HEMTs. The reliability performance was also compared between two vendors of AlGaIn/GaN epilayers. The results indicate that the reliability performance of AlGaIn/GaN HEMTs could strongly depend on the material quality of AlGaIn/GaN epitaxial layers on SiC substrates.

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1. Introduction

AlGaIn/GaN HEMTs have demonstrated excellent power handling capability significantly outperforming state-of-the-art power performance of their counterparts—GaAs and InP based HEMTs. Continuous-wave (CW) power densities of AlGaIn/GaN HEMTs as high as 11.2 W/mm have been demonstrated at 10 GHz [1]. AlGaIn/GaN HEMT X-band power amplifiers achieved CW output power level up to 22.9 [2] and 38 W [3]. The superb power performance of AlGaIn/GaN HEMTs is attributed to the combinations of high breakdown field (3×10^6 V/cm) due to large bandgap, high electron saturation (2×10^7 cm/s) and overshoot velocities (3×10^7 cm/s), and high electron sheet density (1×10^{13} cm^{-2})

[4]. The high electron sheet density is due to strong spontaneous and piezoelectric polarization without intentional doping in AlGaIn–GaN heterostructure. Accordingly, AlGaIn/GaN HEMTs could potentially become the critical components in the areas of satellite communications and high-performance radar applications.

To ensure the reliable operation of AlGaIn/GaN HEMTs for high power amplifiers in a variety of military and space applications, it is essential to demonstrate high reliability performance for the potential insertion of AlGaIn/GaN HEMTs. Recently, most reliability studies of AlGaIn/GaN HEMTs were focused on hot electron stress either under high V_{ds} and/or RF overdrive at room temperature [5–8]. The primary degradation mechanism is attributed to the hot electrons generated in the region between the gate and drain under either high V_{ds} or high RF overdrive stress. The generated hot electrons are likely to be trapped in the interface between the

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passivation layer and AlGaN barrier layer or in the AlGaN barrier layer. The trapped hot electrons could widen the surface depletion region, thus leading to the resultant DC and RF degradations [5]. Furthermore, Si_3N_4 passivation was found to be more resistant to the hot-carrier-induced-degradation (HCID) than SiO_2 passivation under DC and RF overdrive stress [6]. Nevertheless, the reliability investigation of AlGaN/GaN HEMTs under elevated temperature is still lacking. Although DC and RF overdrive stress of AlGaN/GaN HEMTs at room temperature provide reliability information owing to hot electron effect, it is important to explore the reliability of AlGaN/GaN HEMTs subjected to elevated temperature lifetesting in order to warrant the potential insertion of AlGaN/GaN HEMTs for the military and space applications.

In this paper, we investigate the degradation characteristics of AlGaN/GaN HEMTs under elevated temperature lifetesting. It has been found that the degradation mechanism of AlGaN/GaN HEMTs under elevated temperature lifetesting is different from that of GaAs and/or InP HEMTs. Furthermore, the reliability performance of AlGaN/GaN HEMTs between two epitaxial layer vendors was compared. It was found that the reliability performance of AlGaN/GaN HEMTs strongly depends on the material quality.

2. Device technology

The AlGaN/GaN HEMT structure was grown by metalorganic chemical vapor deposition (MOCVD) on 2-in. semi-insulating SiC substrates from vendor A. SiC substrates with a thermal conductivity of 4.9 W/cm K are known to provide better thermal management compared to sapphire substrates which have a thermal conductivity of 0.5 W/cm K. Accordingly, SiC was chosen as the base-line substrates for AlGaN/GaN epi-

layer growth in Northrop Grumman Space Technology (NGST). Fig. 1 shows the device structure used in this study, consisting of a 0.1- μm AlN nucleation layer, a 1- μm -thick GaN buffer layer, and a 250 Å $\text{Al}_{0.28}\text{Ga}_{0.72}\text{N}$ barrier layer. Owing to the large lattice mismatch between AlGaN/GaN and SiC substrates, an AlN nucleation layer was grown to prevent the misfit dislocations induced in SiC–AlN interface from propagating into GaN–AlGaN epilayers. A micrograph (shown in Fig. 2) of scanning-transmission-electron-microscopy (STEM) using a focused-ion-beam (FIB) technique to prepare the cross-section illustrates that misfit locations originated from AlN–SiC were confined in the AlN nucleation layer. However, there are still misfit locations originated from AlN–GaN and extending further into AlGaN epilayer.

The measured Hall sheet carrier concentration and electron mobility are $1.6 \times 10^{13} \text{ cm}^{-2}$ and $1146 \text{ cm}^2/\text{Vs}$, respectively. The AlGaN/GaN HEMT fabrication process is based on NGST's mature GaAs and InP HEMTs technologies designed for high-volume production. The device mesa etch was performed using $\text{BCl}_3/\text{Cl}_2/\text{Ar}$ with inductively coupled plasma (ICP) etching technique. Ti/Pt/Au metal stacks were annealed at 860 °C in a N_2 ambient to form the ohmic contacts with an average contact resistance of 0.65 $\Omega \text{ mm}$. A 0.25 μm T-gate with Pt/Au metal stacks was patterned by two layers (PMMA, P(MMA-MAA)) electron beam lithography followed by PECVD nitride passivation. The details were reported by Smorchkova et al. [9]. The fabricated AlGaN/GaN HEMTs demonstrate the maximum drain current density greater than 1 A/mm at $V_{\text{gs}} = +1 \text{ V}$ and the average peak transconductance greater than 250 mS/mm. The on-state V_{ds} burnout voltage at $I_{\text{ds}} = 500 \text{ mA/mm}$ is as high as 40 V. The cutoff frequency F_T and maximum oscillation frequency F_{max} at V_{ds} of 20 V and I_{ds} of 425 mA/mm are 44 and 80 GHz, respectively. The

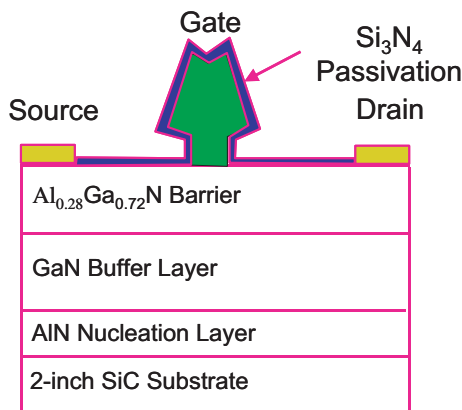


Fig. 1. Cross-section of an AlGaN/GaN HEMT.

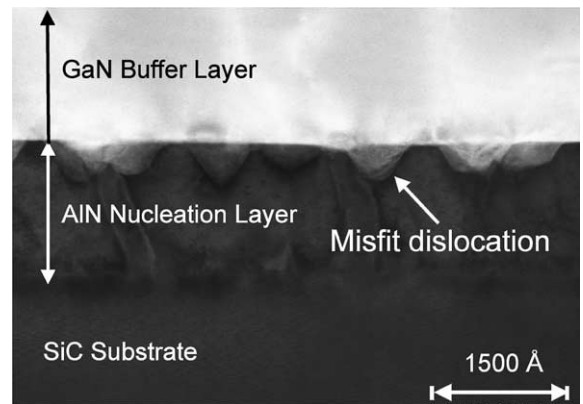


Fig. 2. A STEM micrograph showing the SiC substrate, AlN nucleation layer and GaN buffer layer.

excellent power and noise performance of AlGaIn/GaN HEMTs at frequency of 20 GHz and beyond using NGST's AlGaIn/GaN HEMT technology were described in Refs. [10–12].

3. Lifetest results and discussion

AlGaIn/GaN HEMTs of sample size 10 with four gate fingers and 200 μm gate periphery were mounted on a 16-pin dual-in-line package (DIP) for elevated temperature lifetesting in N_2 ambient, starting from T_a of 150 $^\circ\text{C}$ and ending at T_a of 240 $^\circ\text{C}$ with a step of 15 $^\circ\text{C}$ and a 48 h cycle for each temperature. The devices were stressed at V_{ds} of 10 V and I_{ds} of 500 mA/mm with an estimate junction temperature rise of 150 $^\circ\text{C}$ based on finite-element thermal simulation. The comprehensive DC characteristics of forward/reverse diodes, $I_{ds}-V_{gs}$ and $I_{ds}-V_{ds}$ characteristics were measured on AlGaIn/GaN HEMTs before and after lifetesting. Fig. 3 shows the I_{ds} degradation versus lifetesting ambient temperature. It was observed that device degradation was initiated approximately at ambient temperature of 195 $^\circ\text{C}$. Fig. 4 illustrates that the degradation of I_{max} (measured at $V_{ds} = 5$ V and $V_g = +1$ V) and peak transconductance (G_{mp}) tracks the increase of channel-on-resistance (measured at linear region of $I_{ds}-V_{ds}$ characteristics close to $V_{ds} = 0$ V). Since hot-carrier-induced-degradation has negative temperature coefficient, it is unlikely that the increase of channel-on-resistance observed in our stress conditions is due to the hot-electron trapping effect, typically incurred during the high V_{ds} and/or RF-overdrive stress at room temperature.

The $I_{ds}-V_{ds}$ characteristics on a degraded AlGaIn/GaN HEMT after lifetesting at T_a of 210 $^\circ\text{C}$ are shown in Fig. 5. Devices exhibit I_{ds} and G_m decreases. Fig. 6 shows a typical evolution of $I_{ds}-V_{gs}$ characteristics of AlGaIn/GaN

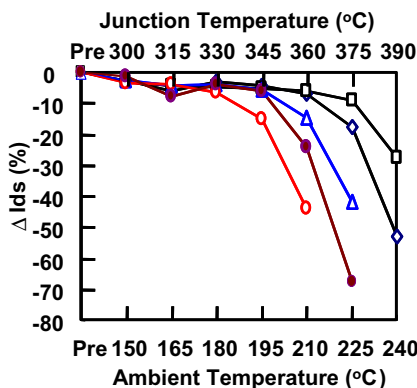


Fig. 3. ΔI_{ds} (measured at $V_{ds} = 5$ V and $V_g = 0$ V) versus lifetesting ambient temperature on AlGaIn/GaN HEMTs grown by vendor A.

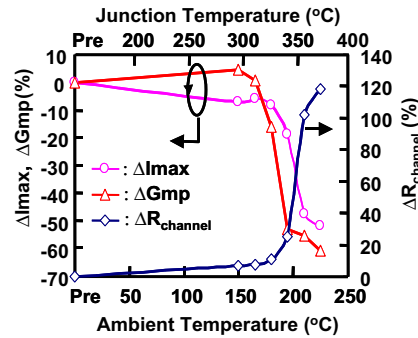


Fig. 4. Correlation of ΔI_{max} , ΔG_{mp} , and $\Delta R_{channel}$ versus lifetesting ambient temperature.

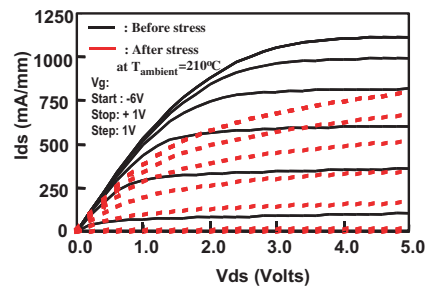


Fig. 5. $I_{ds}-V_{ds}$ characteristics of a degraded AlGaIn/GaN HEMT compared to those of a device before elevated temperature step stress.

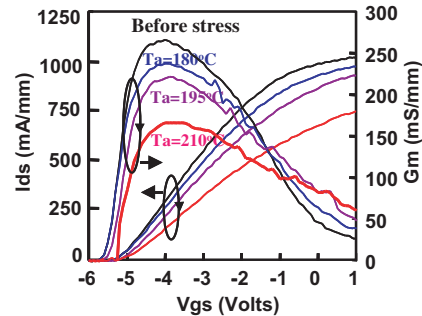


Fig. 6. The evolution of $I_{ds}-V_{gs}$ characteristics of an AlGaIn/GaN HEMT subjected to elevated temperature lifetesting.

HEMT under elevated temperature lifetesting. It was observed that the change of pinch-off voltage is negligible with a slight shift of V_g at peak G_{mp} . In addition, characteristics of gate-drain and gate-source diodes were examined. As shown in Fig. 7, only slight reduction of reverse gate current was observed. The degradation of forward diodes is unnoticeable. This information indicates that the area underneath the gate metal is probably not the origin of degraded AlGaIn/GaN HEMTs under

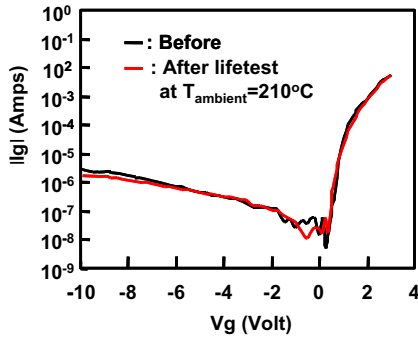


Fig. 7. Reverse and forward diode characteristics of an AlGaIn/GaN HEMT before and after elevated temperature stress.

elevated temperature lifetesting because devices degradation induced by gate sinking exhibit G_m and pinch-off voltage shift as reported by Chou et al. [13].

Moreover, FIB, STEM, and high-resolution energy-dispersive-analysis with X-ray (EDX) were introduced to explore the physical evidence of a degraded AlGaIn/GaN HEMTs. While conventional TEM has long been a standard tool for high-resolution imaging, the STEM has evolved rapidly into a nano-analytical tool while maintaining the atomic resolution imaging capability of TEM. An EDX tool with high-resolution capability is incorporated into the STEM for the chemical analysis if necessary. The combinations of FIB, STEM, and high-resolution EDX, in addition to electrical characterization, provide the powerful capability to resolve the degradation of AlGaIn/GaN HEMTs induced by elevated temperature lifetesting.

This technique was also used for the degradation analysis in GaAs HEMTs under accelerated temperature lifetesting to disclose the complete phenomenon of gate-metal-sinking-induced-degradation (GSID) mechanism in GaAs HEMTs [13]. Fig. 8 shows a STEM micrograph on a degraded AlGaIn/GaN HEMT after lifetesting at

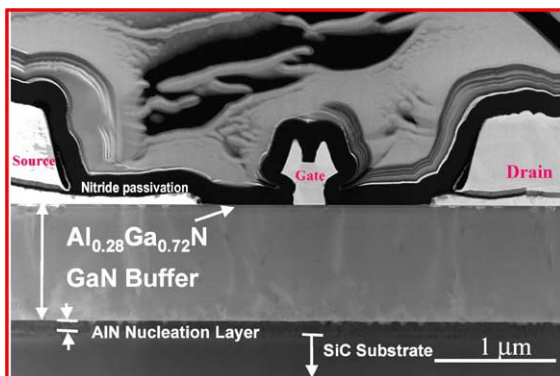


Fig. 8. A STEM micrograph of a degraded AlGaIn/GaN HEMT after elevated temperature lifetesting at ambient temperature of 240 °C for 48 h.

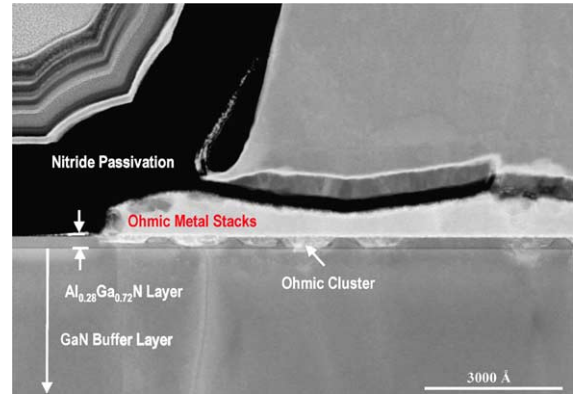


Fig. 9. Close view of a STEM micrograph on the ohmic region of a degraded AlGaIn/GaN HEMT (as shown in Fig. 8).

ambient temperature of 240 °C, exhibiting more than 50% degradation of I_{max} and G_{mp} . A STEM on a virgin device was also performed as the control. It was observed that the misfit dislocations originate from the AlN–GaIn interface and extend up to AlGaIn layer on both virgin and degraded devices. It is still unclear if the misfit dislocations might cause the degradation in our lifetesting. The STEM close view of the ohmic metal region on a degraded device is shown in Fig. 9. There is no signature that ohmic metal interdiffusion is responsible for the degradation of AlGaIn/GaN HEMTs. In addition, as shown in Fig. 9, ohmic clusters were formed during ohmic annealing at 860 °C in order to facilitate the ohmic contact formation. Ohmic clusters were observed on both virgin and degraded devices. Fig. 10 shows a STEM close view of the interface between gate metal and AlGaIn barrier layer on a degraded sample, showing no evidence of gate metal interdiffusion involved in the degradation of AlGaIn/GaN HEMTs induced by elevated temperature lifetesting.

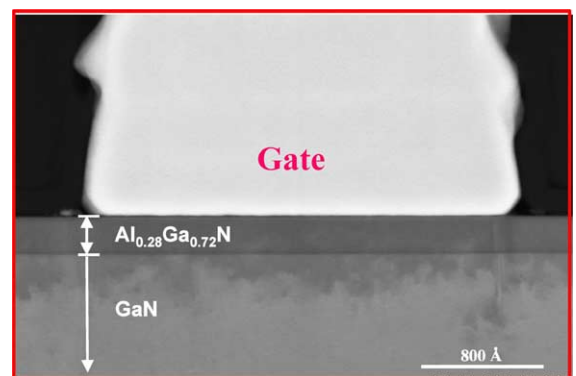


Fig. 10. High magnification of a STEM micrograph on a degraded AlGaIn/GaN HEMT (as shown in Fig. 8). No signature of gate metal sinking was observed.

The STEM results from Figs. 9 and 10 further substantiate the DC evolution in Fig. 6. These are different from the results in GaAs HEMTs induced by elevated temperature lifetesting [14]. What might cause the observed degradation here is still not clear. We speculate that the degradation might be induced by the change of surface charge distribution between the nitride passivation and the AlGaIn barrier layer, ohmic contact resistance degradation of Ti/Al/Pt/Au ohmic metal stacks or imperfect material quality (e.g., misfit dislocation in the epilayers) on AlGaIn/GaN HEMTs subjected to elevated temperature lifetesting. The concrete degradation mechanism is still under investigation.

4. Dependence of reliability on material vendors

Similar lifetesting with sample size of 10 was also performed on AlGaIn/GaN HEMTs fabricated onto the epitaxial layers from vendor B. As shown in Table 1, the devices from vendor B have reverse I_g of $-78 \mu\text{A}$ at V_g of -10 V , which is much higher than the devices fabricated from vendor A with reverse I_g of $-2.16 \mu\text{A}$ at V_g of -10 V , even though the devices from both vendors A and B demonstrate equivalent DC and RF characteristics. However, devices fabricated from vendor B exhibit significant degradation after lifetesting at ambient temperature of $150 \text{ }^\circ\text{C}$ for 48 h, while the degradation at this condition from vendor A's devices is negligible. This result suggests that the reliability performance of AlGaIn/GaN HEMTs could strongly depend on the material quality among different vendors. The material quality could be affected by the variations of epilayer thickness, Al concentration in AlGaIn layer, and defects in the epilayers. As shown in Fig. 8, the misfit dislocations originating from AlN–GaN interface could propagate into AlGaIn barrier layer. Although misfit dislocations were believed to modify the local charge distribution, it is still unclear if the misfit dislocations might affect the reliability performance in AlGaIn/

GaN HEMTs. Based on the reliability performance dependence on different vendors, the improvement of material quality of AlGaIn/GaN HEMT heterostructure on SiC substrate plays an important role for the further improvement of reliability performance in AlGaIn/GaN HEMT technology.

5. Conclusion

Elevated temperature lifetesting was performed on $0.25 \mu\text{m}$ AlGaIn/GaN HEMTs grown by MOCVD on 2-in. SiC substrates. The degradation characteristics consist of a decrease of drain current and transconductance, and an increase of channel-on-resistance accompanying by a negligible shift of pinch-off voltage. There is no detectable ohmic metal or gate metal interdiffusion into the epitaxial materials based on FIB/STEM cross-sectional results. The reliability performance was also compared between two vendors of AlGaIn/GaN epilayers. The results indicate that the reliability performance of AlGaIn/GaN HEMTs could strongly depend on the material quality of AlGaIn/GaN epitaxial layers on SiC substrates.

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Table 1
Reliability performance comparison between vendors A and B

Vendors	I_g (μA) at -10 V	G_{mp} (ms/mm)	I_{max} (mA/mm)
Vendor A before stress	-2.16	260	2065
Vendor A after $T_a = 150 \text{ }^\circ\text{C}$ for 48 h	-2.34	-0.07%	-2.30%
Vendor A After $T_a = 150, 165, 180,$ $190, 210, \text{ and } 225 \text{ }^\circ\text{C}$ (each temperature for 48 h)	-2.14	-57%	-53%
Vendor B before stress	-78.416	263.4	1100
Vendor B after $T_a = 150 \text{ }^\circ\text{C}$ for 48 h	-12.564	-32%	-29

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